



Research Article

ISSN : 0975-7384  
CODEN(USA) : JCPRC5

Preparation and characterization of double perovskite  $\text{LaKFe}_{2-x}\text{Mo}_x\text{O}_6$

Hailang Ju, Baohe Li\*, Jun Gao, Liangting Cai and Yanan Hu

School of Science, Beijing Technology and Business University, Beijing, China

ABSTRACT

The micro-powder of  $\text{LaKFe}_{2-x}\text{Mo}_x\text{O}_6$  ( $x=0.45, 0.55, 0.65$ ) was prepared by the sol-gel technique. After annealing, the XRD spectrum of the samples which annealed at different temperature was tested and the influence on structure of the samples was studied. The samples belong to monoclinic system and the space group is  $P2_1/n$  by crystallographic plane index of diffraction peak of the main phase. The diffraction peaks of  $\text{LaKFe}_{2-x}\text{Mo}_x\text{O}_6$  move to the right with the increased Mo doped in. The lattice parameters and volume decrease as  $x$  rise. The results show that the optimal doping ratio is obtained as  $x$  is 0.45 and the optimal samples of the series is  $\text{LaKFe}_{2-x}\text{Mo}_x\text{O}_6$ . The diffraction peaks of  $\text{LaKFe}_{2-x}\text{Mo}_x\text{O}_6$  move to the right with the rise of temperature and the optimal annealing temperature of it is 1473K.

**Key words:** Double Perovskite, Magnetic properties, colossal magnetoresistance, sol-gel

INTRODUCTION

Crystal structure and physical properties of the perovskite-type manganese oxide is extremely sensitive to their stoichiometric ratio of the constituent elements so minor modulation in composition may change a lot the nature of the compound. So the doping method become an important method in the study of such compounds. The concentration and mobility of the carriers can be improved by doping. Colossal magnetoresistance materials gets great deal of attention because it has a good prospect in the field of electronics, especially in magnetic sensors and high-density magnetic memory. Rare earth manganese oxide become a research focus because they have a wealthy physical meaning due to the colossal magnetoresistance (CMR) [1]. The nature of the substance is directly determined by the structure of matter, so the rich physical properties of perovskite-type oxides is closely related to its own special structure[2-5] and double perovskite oxides is an important part of it. double perovskite oxides have significant applications in superconductivity[6], colossal magnetoresistance room temperature magnetoresistance materials and etc. Previous research has shown that  $\text{Sr}_2\text{FeMoO}_6$  and  $\text{Sr}_2\text{FeReO}_6$  have CMR at room temperature.[7-10] Pradheesh R researched spin glass state in weakly ferromagnetic  $\text{Sr}_2\text{FeCoO}_6$  double perovskite[11],

The double perovskite oxides can expressed as  $\text{A}_2\text{B}'\text{B}''\text{O}_6$  and the schematic picture is shown in Figure 1, wherein A represents an element of the larger ionic radius such as divalent alkaline earth metals Ca, Ba, Sr and Pb and B' or B'' are transition elements such as Fe, Co, Ni and Mn with a smaller ion radius[12].

In the doped oxide the radius of A and B was difference which resulted in crystal structure distortion. It can be measure by the tolerance factor  $t$ [13].

$$R_A + R_O = t(R_B + R_O) \quad (1)$$

The radius of ion A, B and O can be expressed as  $R_A$ ,  $R_B$  and  $R_O$ . Usually if  $t$  is in the range of 0.89 to 1.02 they can form a stable crystal structure. When  $t$  is 1, it corresponds to a full cubic dense structure. With the doping on A or B site, the  $t$ -factor and symmetry of the system will change. Meanwhile, the tolerance factor  $t$  is closely related to temperature and pressure. Usually with the increasing of temperature  $t$  becomes larger and with the decreasing of pressure  $t$  becomes larger.[14]

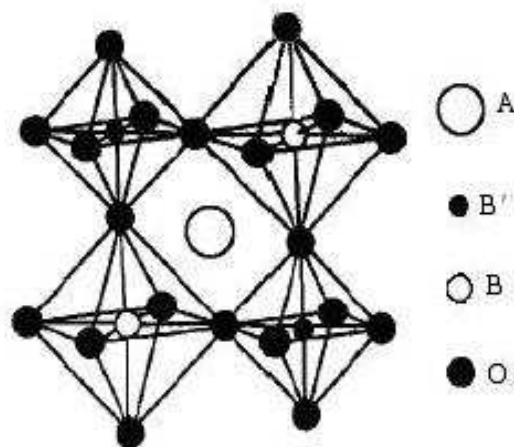


Figure 1 Schematic picture of double perovskite structure  $A_2B'B''O_6$

### Experiment

The micro-powder of  $LaKFe_{2-x}Mo_xO_6$  ( $x=0.45, 0.55, 0.65$ ) were prepared by the sol-gel technique[15]. The crude materials included  $La_2O_3$ (99.99%),  $KNO_3$ (99.99%),  $Fe(NO_3)_3 \cdot 9H_2O$ (AR),  $(NH_4)_2MoO_7$  etc. The experimental procedure is as follows:

- (1) to weigh materials with an electronic balance;
- (2) to configure raw material into a solution;
- (3) to stir and heat the solution and to get black powder sample;
- (4) to calcine the powder in the cased furnace at 1073 K for 2h
- (4) to tablet the calcined the powder with a size of  $\Phi 10\text{ mm} \times 2\text{ mm}$  in the squeezer;
- (5) to anneal the samples at 1473K, 1573K and 1673K for 24 h in the cased furnace and then naturally cool to room temperature in air.

The structure of the samples was determined by X-ray diffraction on a diffractometer of RK D/Max- $\gamma$ A type (Cu  $K\alpha$ ) with rotating target. The diffraction peaks were calibrated.

### RESULTS AND DISCUSSION

Figure 2-4 show the XRD patterns of  $LaKFe_{2-x}Mo_xO_6$  annealed at 1473K, 1573K and 1673K respectively. Figure 5 shows the XRD patterns of  $LaKFe_{1.55}Mo_{0.45}O_6$  annealed at 1473K, 1573K and 1673K.

The samples belong to monoclinic system and the space group is  $P2_1/n$  by crystallographic plane index of diffraction peak of the main phase[16]. The results show that the diffraction peaks move to the right with the increase Mo doped in. As mentioned earlier, the double perovskite oxides is expressed as  $A_2B'B''O_6$ , Fe and Mo is in the position of B' and B'' in this series of experiments. By doping different content of Mo, the valence of it is change as the increase of  $x$ . Gradually  $Mo^{+6}$  became more than  $Mo^{+5}$ , and the ionic radius of  $Mo^{+6}$  is smaller than  $Mo^{+5}$ , so the total ionic radius of B' and B'' become smaller. This leads to the lattice parameters and volume decrease, thus the diffraction peaks move to the right.

In Figure 2, the characteristic peaks in the XRD spectrum of  $LaKFe_{2-x}Mo_xO_6$  is very obvious and the spectrum has almost no impurity one so the crystallinity of the samples is full. In Figure 3, impurity phase appears in the spectrum of  $LaKFe_{2-x}Mo_xO_6$  when  $x$  is 0.55 and 0.45 which was annealed at 1573K. The spectrum of  $LaKFe_{1.55}Mo_{0.45}O_6$  has no impurity one. In Figure 4, the spectrum of three samples have impurity phase all when they were annealed at 1673K. If the annealing temperature is too high, it is not conducive to the crystallization of the samples. Finally, the optimal doping ratio is obtained as  $x$  is 0.45 and the optimal samples of the series is  $LaKFe_{2-x}Mo_xO_6$ .

Through the research above, sample  $LaKFe_{1.55}Mo_{0.45}O_6$  is relatively stable in the high temperature process of annealing. Figure 5 shows the XRD spectrum of  $LaKFe_{1.55}Mo_{0.45}O_6$  annealed at 1473K, 1573K and 1673K. It can be

seen that the crystallinity of  $\text{LaKFe}_{1.55}\text{Mo}_{0.45}\text{O}_6$  is full when the annealing temperature is 1473K and the impurity phase appears when the annealing temperature rise to 1573K and 1673K. The diffraction peaks move to the right with the rise of temperature, this because the higher annealing temperature leads to more fully crystallization so the lattice parameters and volume decrease so the diffraction peaks move to the right. So the optimal annealing temperature of  $\text{LaKFe}_{1.55}\text{Mo}_{0.45}\text{O}_6$  is 1473K.

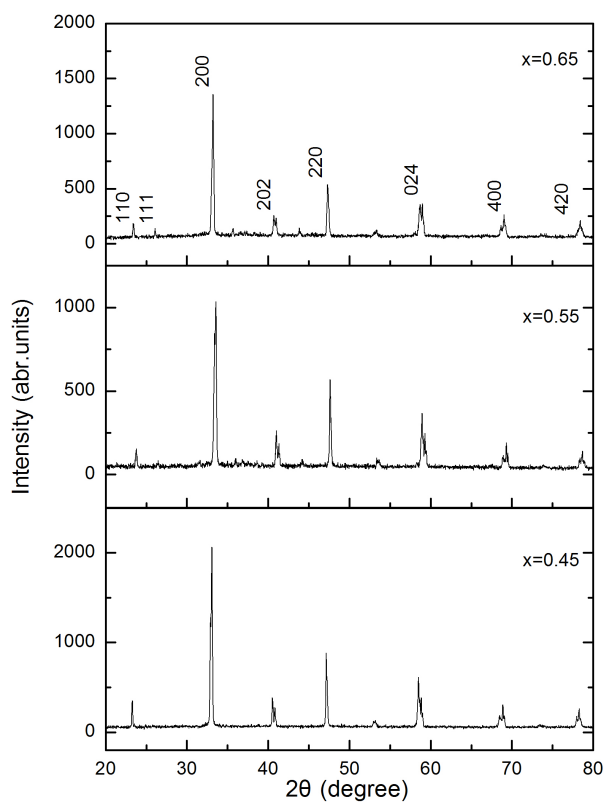


Figure 2 the XRD spectrum of  $\text{LaKFe}_{2-x}\text{Mo}_x\text{O}_6$  annealed at 1473K

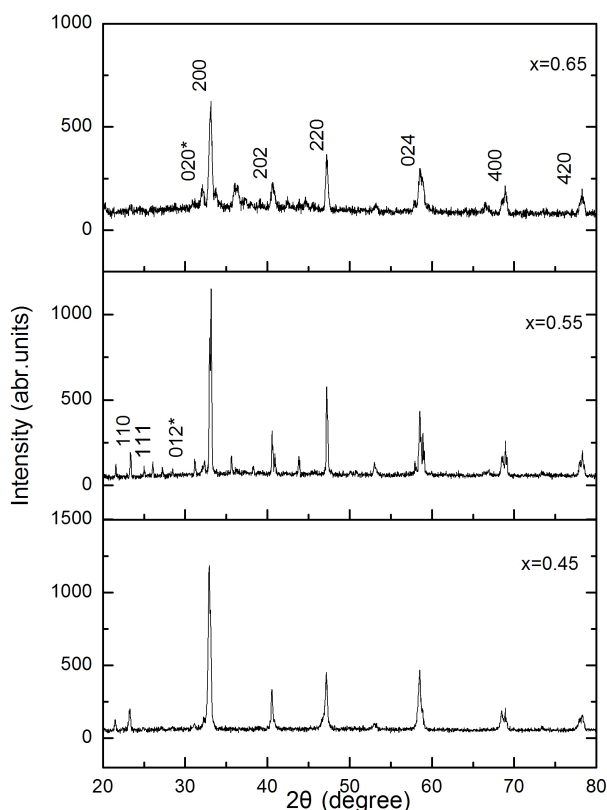


Figure 3 the XRD spectrum of  $\text{LaKFe}_{2-x}\text{Mo}_x\text{O}_6$  annealed at 1573K

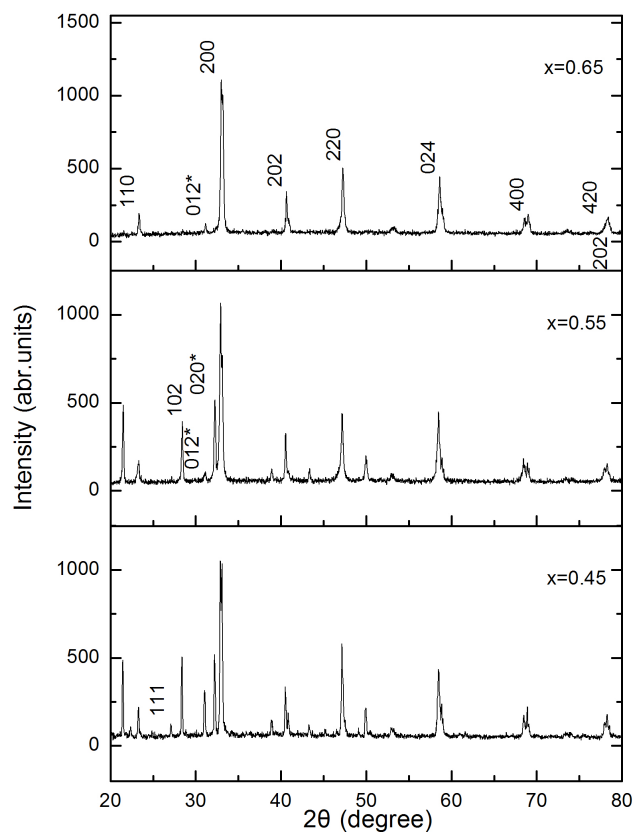


Figure 4 the XRD spectrum of  $\text{LaKFe}_{2-x}\text{Mo}_x\text{O}_6$  annealed at 1573K

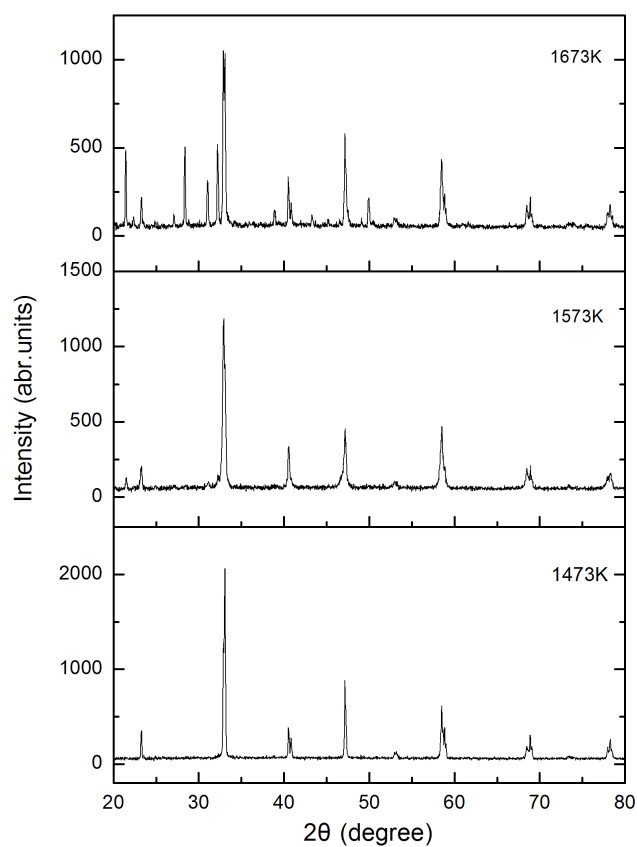


Figure 5 the XRD spectrum of  $\text{LaKFe}_{1.55}\text{Mo}_{0.45}\text{O}_6$  annealed at 1473K, 1573K and 1673K

## CONCLUSION

The micro-powder of  $\text{LaKFe}_{2-x}\text{Mo}_x\text{O}_6$  ( $x=0.45, 0.55, 0.65$ ) were prepared by the sol-gel technique. After annealing treatment, the X ray diffraction tests of different annealing temperature on the influence of the structure of the samples was studied. The optimal doping ratio is obtained as  $x=0.45$  and the optimal samples of the series is obtained as  $\text{LaKFe}_{2-x}\text{Mo}_x\text{O}_6$  and the optimal annealing temperature of it is 1473K.

## Acknowledgments

The authors wish to thank National Natural Science Foundation of China (Grant No. 11174020), and by Teaching masters project of Peking College (Grant No. PXM2013\_014213\_000013), and by the Research Foundation for Youth Scholars of Beijing Technology and Business University (QNJJ2012-33), under which the present work was possible.

## REFERENCES

- [1] Ramirez, A.P., Colossal magnetoresistance. *Journal of Physics: Condensed Matter*, **1997**, 9(39):8171.
- [2] C. N. R. Rao, A. K. Cheetham. Giant magnetoresistance in transition metal oxides[J]. *Science*, **1996**, 272: 369-370
- [3] Arizmendi L. Photo nic Applications of Lithium Niobate Crystals [J]. *Phy. Status Solidi(a)*, **2004**, 201: 253~283
- [4] Ashkin A, Boyd G D, Dziedzic J M, et al. Optically-induced refractive index inhomogeneities in  $\text{LiNbO}_3$  and  $\text{LiTaO}_3$  [J]. *Appl Phys Lett*, **1966**, 9 (1): 72~74
- [5] Volk T, Rubina N, Wohlecke M, Optical-damage-resistant Impurities in Lithium Niobate [J]. *J. Opt. Soc. Am. B*, **1994**, 11(9):1681-1687
- [6] R. J. Cava, B. Batlogg, J. J. Krajewski, et al. Superconducting in a double-perovskite structure [J]. *Nature*, **1988**, 332: 814-815.
- [7] Hanssen K E H M, Mijnders P E, Rabou L P L M, et al. Positron-annihilation half metallic ferromagnet  $\text{NiMnSb}$ : Experiment [J]. *Physical Review B*, **1990**, 42:1533-1540.
- [8] Shirai M, Ogawa T, Kitagawa I, et al. Band structures of  $\text{Zn}$ -blend-type  $\text{MnAs}$  and  $(\text{MnAs})_1(\text{GaAs})_1$  super-lattice. *Journal of Magnetism and Magnetic Materials*, **1998**, 177/181:1383-1384
- [9] Son Y W, Cohen M L, Louie S G. Half-metallic graphene nanoribbons [J]. *Nature*, **2006**, 444:347-349.
- [10] Wu Menghao, Zhang Zhuhua, Zeng Xiaocheng. Charge-injection induced magnetism and half metallicity in single-layer hexagonal group III/V (BN, BP, AlN, AlP) systems [J]. *Applied Physics Letters*, **2010**, 97:093109-093111
- [11] Pradheesh R, Nair H S, Kumar C M N, et al. Observation of spin glass state in weakly ferromagnetic  $\text{Sr}_2\text{FeCoO}_6$  double perovskite [J]. *Journal of Applied Physics*, **2012**, 111(5): 053905-053905-7.
- [12] D. D. Sarma, E. V. Sam Pathkumaran, Sugata Ry, et al. Magnetoresistance in ordered and disordered double perovskite oxides  $\text{Sr}_2\text{FeMoO}_6$  [J]. *Solid State Comm.*, **2000**, 114: 465-68.
- [13] Helmolt R von, Wecker J, et al. Large magnetoresistance and field-induced transition of manganites [J]. *Phys. Rev. Lett.* **1993**, 71:2331-2333
- [14] Y. Moritomo, H. Kuwahara, and Y. Tomioka, Pressure effects on charge-ordering-transitions in Perovskite manganites [J]. *Phys. Rev B*, **1997**, 55, 7549
- [15] Xianyu Wen-xu, Li Bao-he, Qian Zheng-nan, et al. *J. Appl. Phys.* 86(9), (1999) 5164
- [16] Zhang Hongrui, Synthesis and Characterization of Perovskite-K-Fe-Mo-O Series Compound by Sol-Gel Method. Baoding, China. Hebei university, **2009**